Effect of calcination on the structural, optical and magnetic properties of BaWO₄ nanoparticles synthesized by chemical precipitation

Seenamol K Stephen¹, Aloysius Sabu N², Priyanka K P² & Thomas Varghese²*
¹T M Jacob Memorial Government College, Manimalakkunnu, Koothattukulam 686 662, India
²Nanoscience Research Centre (NSRC), Nirmala College, Muvattupuzha 686 661, India

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BaWO₄ nanoparticles have been synthesized by chemical precipitation method using barium nitrate and sodium tungstate. Thermogravimetric analysis has been done to determine the thermal behaviour of the sample. The synthesized nanoparticles have been calcined at 400, 550 and 700 °C for 3 h to get well defined crystalline nanoparticles. As synthesized samples have been characterized by X-ray diffraction, Fourier transform infrared spectroscopy, Raman spectroscopy, scanning electron microscopy, transmission electron microscopy, UV-Visible absorption and photoluminescence spectroscopy. Magnetic properties of the samples have been studied by vibrating sample magnetometer measurements. The effect of calcination temperature on the structural, optical and magnetic properties of BaWO₄ has also been investigated.

Keywords: BaWO₄ nanoparticles, Structural properties, Optical properties, Magnetic properties

1 Introduction

Tungstate nanoparticles have received wide spread attention due to its novel structural, electrical, optical and magnetic properties. They have found many applications in various fields, such as electrical and humidity sensors¹,², oscilloscopes³, light emitting diodes (LEDs)⁴-⁶, gas sensors⁴. Among tungstates, barium tungstate with scheelite structure is important because of its blue luminescence. It has found many applications in the field of fluorescent lamps, optoelectronic devices and display systems⁵. Barium tungstate has been prepared by different methods, such as solid state reaction⁶, microwave irradiation method⁷, template free precipitation technique⁸, co-precipitation method⁹, sucrose templated method¹⁰, modified combustion technique¹¹, sucrose solution evaporation method¹², electrochemical synthesis¹³, catanionic reverse micelles involving polymers¹⁴, sonochemical method¹⁵, microwave hydrothermal synthesis¹⁶, polymeric precursor method¹⁷ and precipitation method¹⁸.

Here we report synthesis and characterization of BaWO₄ nanoparticles by chemical precipitation method without using any surfactant. As synthesized nanoparticles are thermally characterized by thermogravimetric and differential thermogravimetric analysis (TGA/DTG); structurally characterized by X-ray diffraction (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM), Fourier transform infrared spectroscopy (FTIR) and Raman spectroscopy; optically characterized by UV-Visible absorption and photoluminescence (PL) spectroscopy. The magnetic properties of the BaWO₄ nanoparticles are explored using vibrating sample magnetometer (VSM). The effect of calcination temperature on the structural, optical and magnetic properties of the synthesized nanoparticles is also investigated. The results obtained from the systematic investigations revealed that BaWO₄ nanoparticles have potential in varied applications, such as visible light photocatalysis, and optoelectronic devices.

2 Experimental Details

Barium nitrate (Ba(NO₃)₂) (99%, Merck) and sodium tungstate (NaWO₄.2H₂O) (98%, Merck) were used without further purification for the synthesis of BaWO₄.

2.1 Preparation of the sample

0.1 M solution each of barium nitrate and sodium tungstate was prepared by dissolving the desired quantity of them in distilled water. Sodium tungstate solution was added to barium nitrate solution drop by drop, and the mixture solution was then stirred well for about 30 min using a magnetic stirrer at room temperature. The white precipitate formed was
washed well with distilled water several times to remove the unreacted reagents and sodium nitrate formed. The product obtained was dried in a hot air oven at 80 °C for about 8 h. The dried particles were calcined at 400, 550, and 700 °C for 3 h in a muffle furnace. BaWO₄ samples calcined at 400, 550 and 700 °C are denoted as S₁, S₂ and S₃, respectively.

2.2 Characterization technique

Thermogravimetric analysis of the precursor was carried out from 40 to 730 °C at 20 °C/min using Perkins Elmer, Diamond Instrument STA 6000. The structural characterization of the samples were done by X-ray powder diffraction using Brucker AXS D8 advance X-ray diffractometer (λ=1.5406 Å, step size=0.02° and step time 29.1s) with CuKα radiation in 2θ range from 5 to 80°). The surface morphology of the samples was analyzed using scanning electron microscope JOEL model JEK-6390 LV. TEM images were recorded on Joel JEM 2100 at an accelerating voltage of 200 kV. FTIR spectra were recorded using Thermo Nicolet, Avatar 370 in a spectral range 600 to 4000 cm⁻¹. Raman spectra of the samples were taken using confocal Raman microscope with AFM (excitation laser of wavelength 633 nm). Diffuse reflectance spectra of the samples were taken using Schimadzu 2600 UV-Visible spectrophotometer in a wavelength range of 200 to 800 nm. The photoluminescence spectra of the samples were taken using Horiba Fluorolog 3 with Xe-lamp as excitation. The hysteresis loop of the samples were drawn at room temperature by taking measurements with vibrating sample magnetometer (Lakeshore VSM 7410) at room temperature in an applied magnetic field sweeping between ±15000 Oe.

3 Results and Discussion

3.1 TG/DTG analysis

Figure 1 shows the thermal decomposition result of the precursor from the ambient temperature to 700 °C using both the thermogravimetric and differential thermogravimetric curves. Only 3% weight loss occurs in the temperature range from 100 to 700 °C. This is due to the loss of surface water on the particles, and combustion of minor impurities present in the sample. The exothermic curve of DTG confirms the combustion of the minor impurities in the sample. DTG curve remains almost constant, and only a small dip is observed at 602 °C due to the faster combustion of the impurity. The thermal analysis shows that the BaWO₄ nanoparticles are thermally stable above 300 °C. Hence, temperatures 400, 550 and 700 °C have been selected for calcination treatment.

3.2 XRD analysis

The phase purity, crystallinity and structure of powder samples were examined using XRD. Figure 2 shows XRD patterns of the BaWO₄ samples calcined at different temperatures. The well-defined diffraction peaks are obtained for the planes (101), (112), (004), (200), (211), (204), (220), (116), (132), (224), (400), (208), (316), (413), (404), (240), (228) and (110), which correspond to tetrahedral crystal structure with a space group I₄₁/a for BaWO₄ (JCPDS file No. 72-0746 of BaWO₄). No impurity peaks were detected in the spectra. Therefore, the present chemical precipitation method is a successful method for the synthesis of BaWO₄ nanoparticles. Lattice parameters of the samples were calculated using plane spacing equation for tetragonal structure

\[ D = k\lambda/\beta \cos \theta, \]  

… (1)
where $\lambda$ is the wavelength of light used, $\beta$ the full width half maximum of the sample, $\theta$ the angle of diffraction and $k$ is a constant equal to 0.9. The crystallite sizes estimated for samples S$_1$, S$_2$ and S$_3$ are 28.58, 32.17 and 37.20 nm, respectively. The micro-strain or micro-diffraction values obtained from Williamson-Hall analysis$^{22}$ are 0.00146, 0.00101 and 0.000767, respectively, for S$_1$, S$_2$, and S$_3$.

The micro-strain of the samples is due to the presence of O vacancies, structural imperfections and surface defects, which in turn results XRD line broadening. As the calcination temperature increases the structural defects probably reduced, and hence the micro diffraction decreases. In short, calcination temperature influences the geometrical parameters of BaWO$_4$ nanoparticles.

### 3.3 Energy dispersive X-ray spectroscopy (EDS)

EDS is an analytical technique for the elemental analysis or chemical characterization of the as prepared sample. The EDS spectrum of the BaWO$_4$ nanoparticles calcined at 400 °C is shown in Fig. 3. It clearly shows the co-existence of barium, tungsten and oxygen in the sample in stoichiometric ratios. Absence of other elemental peaks in the spectrum exhibits the purity of the prepared BaWO$_4$ nanoparticles. The elemental composition of BaWO$_4$ sample is presented in Table 2.

### Table 1 – Geometric parameters of BaWO$_4$ samples.

<table>
<thead>
<tr>
<th>Sample</th>
<th>$a = b$ (Å)</th>
<th>$c$ (Å)</th>
<th>Unit cell volume (Å$^3$)</th>
<th>Crystallite size (nm)</th>
<th>Micro strain</th>
</tr>
</thead>
<tbody>
<tr>
<td>S$_1$</td>
<td>5.5894</td>
<td>12.658</td>
<td>395.45</td>
<td>28.578</td>
<td>0.00146</td>
</tr>
<tr>
<td>S$_2$</td>
<td>5.6246</td>
<td>12.757</td>
<td>403.58</td>
<td>32.167</td>
<td>0.00101</td>
</tr>
<tr>
<td>S$_3$</td>
<td>5.6160</td>
<td>12.730</td>
<td>401.52</td>
<td>37.206</td>
<td>0.00076</td>
</tr>
</tbody>
</table>

### Table 2 – Elemental composition of BaWO$_4$ sample.

<table>
<thead>
<tr>
<th>Element</th>
<th>Mass %</th>
<th>Atom %</th>
</tr>
</thead>
<tbody>
<tr>
<td>W</td>
<td>51.03</td>
<td>16.53</td>
</tr>
<tr>
<td>Ba</td>
<td>30.05</td>
<td>13.03</td>
</tr>
<tr>
<td>O</td>
<td>18.92</td>
<td>70.43</td>
</tr>
<tr>
<td>Total</td>
<td>100</td>
<td>100</td>
</tr>
</tbody>
</table>

### 3.4 SEM analysis

For taking the SEM micrograph, a small amount of BaWO$_4$ powder is dispersed in ethanol. A drop of the suspension was placed on the carbon tape attached on the stub. It is then kept for about 5 to 10 min under infrared lamp for drying before scanning for micrographs. Figure 4 shows the SEM images of BaWO$_4$ nanoparticles calcined at 400 (S$_1$), 550 (S$_2$) and 700 °C (S$_3$).

The SEM images represent the surface morphology of the samples. It can be seen from Fig. 4 that the morphology of the samples changes with calcination temperature. The figure shows that sample S$_1$ is granular like, S$_2$ is popcorn shaped and S$_3$ has flower like morphology. As the temperature increases agglomeration of the particles takes place. The crystallinity of the sample is also increased as calcinations temperature increased. It can be seen that surface of the particles is roughly porous. As the calcination temperature increases the surface
becomes smooth. The porosity of nanomaterials greatly influences the quality and utility of solid phase chemicals including agrochemicals, additives and pharmaceutical ingredients. It plays major roles in the purification, processing and blending of chemical products. In short, porous materials can help to design successful product and process development.

### 3.5 TEM analysis

TEM images of the samples have been recorded for analyzing the morphology and size of the synthesized products. For recording the TEM, BaWO₄ nanoparticle samples were initially dispersed in ethanol. A drop of the suspension was placed on a carbon coated copper grid and analyzed using a JEOL, JEM 2100 operated at 200 kV. Figure 5 shows the TEM bright field images and selected area electron diffraction (SAED) patterns of the samples calcined at different temperatures. It is found that the particles are almost spherical in shape, and agglomerated. Strong bright spots in the SAED pattern represent the crystalline nature of the sample. These bright spots are produced by the electrons reflected and diffracted from the different crystallographic planes of the unit cells of BaWO₄.

### 3.6 FTIR analysis

The FTIR spectra of the BaWO₄ samples S₁, S₂ and S₃ were recorded in the transmittance mode and are shown in Fig. 6. The spectra have several significant peaks recorded in the range of 4000 to 400 cm⁻¹. IR active modes of BaWO₄ samples are given in Table 3.

Since BaWO₄ has tetrahedral symmetry, F2(v3,v4) modes are infrared active only. The absorption peak at 403 cm⁻¹ can be assigned to F2(v4) vibration, which represents the bending vibration W-O. The anti-symmetric stretching vibration F2(v3) originating from the W-O in the WO₄²⁻ tetrahedron corresponds to the broad intense peak centered at 820 cm⁻¹. The broad absorption band centered at 3450 cm⁻¹ and the sharp peak centered at 1430 cm⁻¹ are assigned to O-H stretching vibrations and the O-H bending vibrations, respectively. The band centered at 1630 cm⁻¹ is attributable to H-O-H bending vibration mode. In brief, the direct precipitation method used in the present work is suitable synthesis route to obtain the tetrahedral BaWO₄ nanoparticles with the characteristics bonds.

### 3.7 Raman studies

Raman spectra of the samples were taken to verify the rotational, vibrational and other low
frequency modes of the system. From group theory
the vibrational representations at the \( \Gamma \) point scale for
scheelite BaWO\(_4\) is given by:

\[
\Gamma = (3A_g+3B_u) + (5B_g+5A_u) + (5E_g+5E_u), \quad \ldots \ (2)
\]

Where \( A \) and \( B \) modes are degenerate, and \( E \) modes
are non-degenerate. The sub-index \( g \) and \( u \) stand for
even and odd parity. One \( A_u \) and one \( E_u \) correspond
to the zero frequency acoustic modes and the rest are
optic modes. In BaWO\(_4\), the first member of pairs is
Raman active. The second member of pairs is infrared
active except for the Ba silent modes that are not
infrared active. There will be 13 Raman active modes
for BaWO\(_4\),

\[
\Gamma = 3A_g + 5B_g + 5E_u. \quad \ldots \ (3)
\]

The Raman spectra of scheelite BaWO\(_4\) contain
internal and external mode of vibration. In the internal
mode of vibration, the centre of mass does not move
and the vibrations are within the \([\text{WO}_4]^{2-}\) molecule.
The external mode of vibration corresponds to the
motion of \( \text{Ba}^+ \) cation and the rigid molecular unit\(^{12,13}\).
Figure 7 depicts Raman spectra of the samples \( S_1 \), \( S_2 \)
and \( S_3 \).

Table 4 compares Raman active modes of BaWO\(_4\)
samples with literature values. Here \( \nu_1 \), \( \nu_2 \), \( \nu_3 \) and \( \nu_4 \)
represent the internal mode of vibrations, \( R \) and \( T \)
represent the rotational and translational mode of
vibrations. It is found that maximum shift is observed
for Raman vibration modes for sample \( S_2 \), which has
the highest unit cell volume. It can be seen that the
experimental values observed are in agreement
with the reported values. Two additional peaks are
also obtained at 754 cm\(^{-1}\) and 891 cm\(^{-1}\) for samples \( S_1 \)
and \( S_2 \). This may be due to impurities present in the
sample. But the XRD, EDAX and FTIR results
excluded the presence of impurities. Hence we can
conclude that the absorption peak at 891 cm\(^{-1}\) is due to
the symmetric stretching vibration of the short \( W-O \)
bond in the \( \text{WO}_2 \) group and the absorption at 754 cm\(^{-1}\)
is due to the surface disordered layers or due to the
strain present in the prepared sample\(^{26}\). The intensity
and FWHM of the Raman modes decrease as the
calcinations temperature increases, which is due
to the change in surface morphology of the samples.
As can be seen from the SEM photographs, the
surface becomes smooth as calcinations temperature
increases. Hence, the absorption as well as emission
decreases or the particles become more crystalline
with calcination.

3.8 UV-Visible absorption studies

UV-Visible absorption spectra of the samples
were taken to analyze their optical absorption
properties. Figure 8 shows the optical absorption
spectra of calcined samples of BaWO\(_4\). The optical
absorption is observed only in the ultraviolet region of
the spectra. The absorption peak corresponding to the
samples S₁, S₂, and S₃ are at 220, 224 and 223 nm, respectively. The small red shift with increase in calcinations temperature is due to increase in particle size caused by the calcination.

Tauc plot of the BaWO₄ samples calcined at different temperatures is depicted in Fig. 9. Tauc relationship was used to calculate the bandgap of the nanoparticles. The bandgap values estimated for S₁, S₂ and S₃ are 5.25, 5.14 and 5.13 eV, respectively. These values are slightly greater than the reported value 4.6 eV. The UV-Visible absorption studies show that optical bandgap decreases with increase in calcination temperature due to crystal growth caused by calcination.

### 3.9 Photoluminescence studies

The PL spectra of the BaWO₄ samples with an excitation wavelength of 388 nm are shown in Fig. 10. One intense peak is observed at 428 nm and two weak peaks at 531 and 640 nm. The blue emission (428 nm) is due to the charge transfer transitions between the O2p orbits and the empty d orbits of the central W⁶⁺ ions in [WO₄]²⁻ tetrahedron. The other two peaks are due to structural distortions in the crystal lattice. Here the emission peaks are sharp, since calcination reduces the defects in the crystal structure and the crystals become highly ordered. The PL emission is found to be in visible regime for all the samples, and the intensity of emission is found maximum for the sample S₃. Thus, PL emission can also be tuned by changing calcination temperature. In short, the PL studies suggested that the synthesized BaWO₄ nanoparticle samples are useful for photocatalytic applications.

#### 3.9.1 CIE chromaticity

The CIE serves as a standard reference against which many other colour spaces are defined. The chromaticity of a colour is designated by projection co-ordinates x and y. The CIE chromaticity co-ordinates under 388 nm excitation of the BaWO₄ samples S₁, S₂ and S₃ are (0.22364, 0.251694), (0.242682, 0.29394) and (0.243769, 0.296648) respectively. Figure 11 shows the CIE chromaticity diagram for the samples S₁, S₂ and S₃. All the three samples show blue-green emission with slight changes for the chromaticity co-ordinates. The CIE results show that BaWO₄ phosphors can be used in fluorescent lamps, electronic display and other optoelectronic applications.

### Table 4 – Comparison of Raman active modes of BaWO₄ samples with literature values.

<table>
<thead>
<tr>
<th>Peak/mode</th>
<th>Theo.¹</th>
<th>Present values</th>
<th>Literature values</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td></td>
<td>S₁</td>
<td>S₂</td>
</tr>
<tr>
<td>T(Bg)</td>
<td>55</td>
<td>-</td>
<td>55</td>
</tr>
<tr>
<td>T(Eg)</td>
<td>81</td>
<td>75</td>
<td>81</td>
</tr>
<tr>
<td>T(Èg)</td>
<td>110</td>
<td>101.7</td>
<td>110</td>
</tr>
<tr>
<td>T(Bg)</td>
<td>145</td>
<td>150.6</td>
<td>145</td>
</tr>
<tr>
<td>R(Èg)</td>
<td>149</td>
<td>-</td>
<td>149</td>
</tr>
<tr>
<td>v2(Èg)</td>
<td>209</td>
<td>191</td>
<td>209</td>
</tr>
<tr>
<td>v2(Bg)</td>
<td>328</td>
<td>332</td>
<td>328</td>
</tr>
<tr>
<td>v4(Èg)</td>
<td>339</td>
<td>345.6</td>
<td>339</td>
</tr>
<tr>
<td>v4(Bg)</td>
<td>348</td>
<td>354.9</td>
<td>348</td>
</tr>
<tr>
<td>v3(Èg)</td>
<td>797</td>
<td>796</td>
<td>797</td>
</tr>
<tr>
<td>v3(Bg)</td>
<td>823</td>
<td>831.6</td>
<td>823</td>
</tr>
<tr>
<td>v1(Èg)</td>
<td>928</td>
<td>928</td>
<td>928</td>
</tr>
</tbody>
</table>

Fig. 8 – UV-Visible absorption spectra of BaWO₄ samples.
3.10 Magnetic properties

The magnetic properties of the BaWO₄ nanoparticle samples were studied using VSM measurements. The hysteresis curves for the samples S₁, S₂ and S₃ are shown in Fig. 12. It can be seen that the samples show ferromagnetic behavior. The saturation magnetization values are very low because of the absence of unpaired electrons in barium, and it agrees with the literature values¹⁸. The variation of coercivity, saturation magnetization, retentivity and squareness ratio $R$ of the samples are presented in the Table 5. Fig. 12 and Table 5 confirm that saturation magnetization and coercivity of the BaWO₄ samples varies with calcinations temperature.
The saturation magnetization is maximum for the sample calcined at 550 °C (S2), which has the highest value of lattice parameters. It is interesting that the change in saturation magnetization and retentivity are in accordance with the change in lattice parameters of the sample with calcinations temperature. The saturation magnetization is found to be maximum for the sample (S2) with maximum unit cell volume. Since the squareness ratio is less than 0.5 we can conclude that these are soft magnetic particles. The coercivity and the squareness ratio are found to increase with increase in calcination temperature. These results confirm that as synthesized BaWO4 nanoparticles are single domain particles. As the calcination temperature increases the size of the particles increases. The coercivity of the single domain particle increases progressively with increase in particle size, because the magnetic moment of the individual particle increases. The magnetic moment of the individual particle increases with increase in particle size and hence, the retentivity of the particles increases. However, the exchange hardening of the soft magnetic particles begin to fail with further increase in particle size and hence the retentivity decreases.

### 4 Conclusions

BaWO4 nanocrystalline samples are successfully prepared by direct chemical precipitation method without using any surfactant, templates or catalyst. From thermal studies, it is found that the synthesized BaWO4 nanoparticles are thermally stable above 300 °C. The XRD spectral studies confirm tetrahedral structure for the synthesized BaWO4 samples with a space group I41/a. The structural characterization confirms changes of particle size and surface morphology of BaWO4 nanoparticles with increase in calcination temperature. The EDS study confirms the purity and elemental composition of the synthesized BaWO4 sample. The optical bandgap energies of the BaWO4 samples are also slightly changed with calcination treatment due to change in particle size caused by calcination. The PL spectra and CIE chromaticity diagram confirm the use of BaWO4 nanocrystalline phosphors for fluorescent applications. From the VSM studies it is found that the synthesized BaWO4 nanoparticles possess ferromagnetic behaviour at room temperature, and there is an anomalous variation of magnetic properties with calcinations temperature. Moreover, the synthesized BaWO4 nanoparticle samples can be used in potential applications such as, optoelectronic devices and fluorescent and photocatalytic applications.

### Acknowledgment

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### References


### Table 5 – Comparison of magnetic properties of the samples.

<table>
<thead>
<tr>
<th>Sample</th>
<th>Saturation magnetization (Ms)</th>
<th>Coercivity (Oe)</th>
<th>Retentivity (Mr)</th>
<th>Squareness ratio $R=Mr/Ms$</th>
</tr>
</thead>
<tbody>
<tr>
<td>S1</td>
<td>$3.229 \times 10^{-3}$ emu/g</td>
<td>302.44</td>
<td>$57.69 \times 10^{-6}$</td>
<td>0.0178</td>
</tr>
<tr>
<td>S2</td>
<td>$5.783 \times 10^{-3}$ emu/g</td>
<td>361.63</td>
<td>$455.05 \times 10^{-6}$</td>
<td>0.0787</td>
</tr>
<tr>
<td>S3</td>
<td>$3.4965 \times 10^{-3}$ emu/g</td>
<td>407.04</td>
<td>$388.2 \times 10^{-6}$</td>
<td>0.111</td>
</tr>
</tbody>
</table>