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Influence of Thermal Annealing to Material Properties of TiO₂, ZnO, and TiO₂/ZnO Bilayer Films by Sol-Gel Method

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This study reported on the material properties of TiO₂, ZnO, and TiO₂/ZnO bilayer thin films fabricated by sol-gel spin coating technique after annealing. ZnO thin films were prepared by dissolving zinc acetate dehydrated into a solvent of ethanol and diethanolamine (DEA) was later added. Meanwhile, for TiO₂ films, titanium tetraisopropoxide (TTIP) was dissolved into ethanol and an acetate acid was added. Afterwards all film samples were annealed at different temperatures in the range of 400 – 600 °C for 60 minutes. For TiO₂/ZnO bilayer thin films, the structural properties showed that after annealing at 600 °C, the X-RD peak intensity was more pronounced compared to at 400 and 500 °C. This indicated that the amorphous phase of the films decreased after annealing at 600 °C. Furthermore, the crystallite sizes of the films increase as the annealing temperature increased to 600 °C, while the strain reduces at this temperature. In the case of (UV-Vis) spectroscopy, it was seen that after annealing the transmittance value of the TiO₂/ZnO bilayer increase. The band gap energy of the films consists of low and high energy was around 3.60 eV corresponding to the impurity transitions. In addition, the refractive index (*n*), the extinction coefficient (*k*), and the real and imaginary part of dielectric constant were determined from the transmittance and absorbance spectra results. Based on the SEM results indicate that there was no cluster in the surface of TiO₂/ZnO bilayer thin films.

Keywords: Sol-gel method, Thermal annealing, TiO2/ZnO bilayer thin films

1 Introduction

The Combination of two semiconductor heterostructures can increase the optical properties and the device performance of the semiconductors rather than a single semiconductor. In this study, ZnO and TiO₂ are the two metal-oxide semiconductors combined. ZnO is one of the II-VI compound semiconductors that have been received much attention due its outstanding optical and electrical properties. ZnO has a band gap energy of 3.3 $eV^{1,2}$ and a large exicton binding energy of 60 meV^{3,4} which is widely used in thin film sensors⁵, light emitting diodes $(LED)^6$ and photovoltaic devices⁷. In addition, it is a good candidate for photoanode in dye sensitized solar cells (DSSCs) application due to its low cost and non-toxicity. However, it has low energy conversion efficiency in photovoltaic performance. Therefore, to increase its device performance as the photoanode in DSSC application, TiO₂ and ZnO were combined, forming the ZnO/TiO₂ bilayer films which produce better photoanode properties.

On the other hand, TiO₂ material has wide band gap energy of 3.2 eV,8 which is widely used in photo catalysts⁹, applied as gas sensing in materials¹⁰ and anti-reflection coatings¹¹. It was also developed as a photoanode material because of its high performance in energy conversion efficiency. Therefore, it was used to improve the properties of ZnO and increase the performance of photoanode in DSSCs application. TiO₂ is classified into three phases of crystalline: anatase, rutile, and brookite^{12,13}. The Rutile phase is the most stable phase and is usually obtained after annealing at temperature above 500 °C¹². The optical band gap of TiO_2 is 3.0 and 3.2 eV for rutile and anatase crystalline phase respectively¹³. Mixed metal oxide TiO2-ZnO nanostructures have been widely used as sensing gas materials¹⁴ and the solar blind photoconductor¹⁵.

There are many techniques that have been used to grow ZnO and TiO₂ thin film; radio frequency (RF) magnetron sputtering^{16,17}, molecular beam epitaxy (MBE)¹⁸, spray pyrolysis^{19,20} and sol gel spin coating method²¹⁻²³. Among other methods, the sol-gel spin

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coating method is one of the most attractive techniques used to fabricate the thin films due to its simplicity, inexpensive cost to characterize the structural and optical properties and the easy control of chemical compound.

In the present study, the structural and optical properties of TiO₂, ZnO, and TiO₂/ZnO bilayer thin film after annealing at different temperature are investigated. To achieve this, we first grow ZnO films on the glass substrates followed by TiO₂ on the top of ZnO thin film using sol-gel spin coating. The crystallite size and the strain of the films were estimated based on the X-ray diffraction results. Furthermore, the UV-Vis spectrophotometer was carried out to estimate the band gap energy of the films. The refractive index (n), the extinction coefficient, and the real and imaginary part of dielectric constant were also characterized based on the transmittance and absorbance results. The surface morphology of the films was observed using the scanning electron microscopy (SEM), and from the SEM results we estimate the thickness of the films.

2 Experimental detail

2.1 Synthesis of ZnO, TiO₂, and TiO₂/ZnO bilayer films

ZnO thin film was prepared using zinc acetate dehydrate (Zn(CH₃COO)₂.2H2O) as a starting material. First, zinc acetate dehydrate was dissolved into 1.5M of ethanol and diethanolamine (DEA) as a solvent and stabilizer, respectively. The solvent was stirred on the magnetic hotplate for 30 minutes and the sol-gel was growth on glass substrate with the speed rate of 3000 rpm for 30 seconds. Next, the films were heated on the hot plate for 30 minutes to evaporate the mixture and unwanted organic residuals. After fabricating the ZnO films as a first layer, the TiO₂ films were fabricated subsequent on the top of ZnO films. 1.5 ml titanium tetraisopropoxide (TTIP) was dissolved into 10 ml ethanol and 0.1 ml acetate acid was added. These solvent were stirred on the magnetic hotplate at 80 °C for 5 hours. Afterwards, the sol-gel was coated on a glass substrate by spin coating with a speed rate of 3000 rpm for 30 seconds and was heated on the hotplate for 30 minutes to remove the undesireable mixture from the surface.

Prior to the fabrication of ZnO/TiO_2 bilayer thin films on the glass substrate, it was cleaned with HCL to remove the unwanted particles from its surface. The ZnO solution was first deposited on the glass substrate to form the ZnO film. Afterwards, it was heated on the hotplate at 300 °C to evaporate the solvent from the surface. Furthermore, the second layer of TiO₂ was deposited on the ZnO films to form the ZnO/TiO₂ bilayer thin films. These layers were heated on the hotplate to evaporate the remaining residual in the films. The ZnO/TiO₂ bilayer films were finally inserted into the furnace to anneal the samples at different temperatures from 400 to 600 °C for 60 minutes.

2.2. Materials characterization

X-ray diffraction (X-RD) measurement was recorded to study the crystallite size and the strain of ZnO, TiO₂ and TiO₂/ZnO using Cu K α radiation. The recorded of diffraction angle was varied from 20° to 70°. The voltage and the current of X-RD were kept a constant at 40 kV and 30 mA, respectively with the speed rate of X-RD was 2°/min. The crystallite size and the strain of the films were estimated based on the X-RD results using Debye-Scherrer equation and William-Hall methods. We used SEM to evaluate the surface morphology of the thin films. Furthermore, the UV-Vis optical transmittance spectroscopy was carried out to estimate the band gap energy of single layer ZnO, TiO₂ and TiO₂/ZnO bilayer thin films using a single beam UV-Vis spectroscopy with a wavelength range of 300 - 800 nm. Further investigation of optical properties such as the refractive index (n), the extinction coefficient (k) and the dielectric were determined using the UV-Vis results.

3 Results and Discussion

3.1 Structural characterization

Figure 1 (a,b) depict the recorded X-RD spectra of a single layer TiO₂ and ZnO film after annealing at different temperatures from 300 to 600 °C for 60 minutes. For TiO₂ thin films, The X-ray diffraction patterns were clearly seen after annealing the films at 300 °C. The diffraction angle of 20 at the X-RD spectra of TiO₂ films were observed at 25.41, 37.97 and 48.31° that corresponded to the reflection of (101), (004) and (200), respectively. After annealing the films at 600 °C, the 20 was shifted to the right at 25.61, 38.08 and 48.51°, respectively. The strong peak was located at 20=25.41° showing an anatase phase with preferential orientation of (101) plane (JCPDS card no 21-1272). Furthermore, previous studies on TiO₂ films showed that the preferential growth of



Fig. 1 — (a) X-RD patterns of TiO₂ thin films after annealing with different temperatures from 300 °C to 600 °C for 60 minutes and (b) shows the X-RD patterns of ZnO thin films after annealing with different temperatures from 300 to 600 °C for 60 minutes.

TiO₂ was in (101) plane direction and was categorized as anatase crystalline phase which was confirmed in the results obtained^{24,25}. Conversely, the recorded X-RD spectra of ZnO films Fig 1(b) annealing at various temperatures revealed four well defined X-RD maxima at diffraction angle of 31.62, 34.72, 36.27 and 47.57° (JCPDS card No: 96-230-0113) that were attributed to reflections plane of (100), (002), (101) and (012), respectively. These crystal planes corresponded to the hexagonal wurtzite phase. In addition, the preferential growth of ZnO films were in (002) direction since its intensity was higher than that of other plane. This result was in agreement with the previous studies where the preferential growth of ZnO films was in (002) direction^{5,26}.

Figure 2 shows the X-RD curves of TiO₂/ZnO bilayer thin films after annealing at different temperatures from 400 to 600 °C. As shown in this figure, after annealing at 400 °C, the intensity of recorded X-RD peak is very weak. However, after annealing at 600 °C the X-RD peak of TiO₂/ZnO bilayer thin film was clearly observed with higher intensity than that 400 and 500 °C, more pronounced X-RD peaks appeared at this annealing temperature. Furthermore, the diffraction angle 2θ of 26.61° , 48.18° assigned to (101) and (200) plane belonged to the TiO₂ anatase phase. Meanwhile, ZnO X-RD peaks were observed at 20=32.51°, 37.09°, 56.84°, 63.15° and 68.38° assigned to the planes of (100), (002), (101), (110), (103), and (201), respectively. A very strong X-RD peak in ZnO film was observed at the diffraction angle of 35.13° represented by (002) plane. In addition, it was discovered that the diffraction angle of 56.84, 63.15 and 68.38° were also represented by the TiO_2 rutile phase of (220), (130) and (112) planes (JCPDS card no 21-1272). It is



Fig. 2 — X-RD spectra of TiO_2/ZnO bilayer thin films after annealing at different temperatures from 400°C to 600°C for 60 minutes. *Note*. A symbol refers to TiO_2 anatase phase.

interesting to note that the X-RD peak related to (004) plane of TiO₂ anatase phase which appeared in the single layer of TiO₂ were not detected in bilayer TiO₂/ZnO bilayer thin films indicating the reduction of TiO₂ phase fraction in this layer. Similar results have also been reported in the previous study by Kheiri *et al.*²⁷ and Choi *et al.*²⁸. They showed that

there was a significant reduction in the TiO_2 fraction when the layers of ZnO were deposited on TiO₂ as a result of the instability in the growth of the TiO₂ layer on ZnO. It was observed that there was an improvement in the crystalline bilayer TiO₂/ZnO bilayer thin films after annealing at 600 °C which was indicated by an increase in the X-RD intensity. The results confirmed in the previous studies in which the intensity of TiO₂/ZnO bilayer thin films after annealing at high temperature was inversely proportional to the amorphous phase^{29,30}.

The crystallite size of the bilayer TiO₂/ZnO bilayer thin films can be estimated using the Debye-Scherrer equation as follows³¹,

$$D = \frac{0.9\lambda}{\beta_{hkl}cos\theta} \qquad \dots (1)$$

Where λ is the X-ray wavelength ($\lambda = 1.5406$ Å), β_{hkl} is the full width at half-maximum (FWHM) and θ is the Bragg angle, respectively. In addition, William-Hall method (Uniform Deformation Method=UDM) was used to calculate the crystallite size and the strain of TiO₂/ZnO bilayer thin films in the following equation^{32,33}.

$$\beta_{hkl}\cos\theta = \frac{k\alpha}{D} + (4\varepsilon\sin\theta) \qquad \dots (2)$$

It is assumed in the equation (2), that the strain is uniform. The fitted line of $\beta_{hkl} \cos\theta$ to $4\sin\theta$ give results the slope and y-intercept that represents to the strain and the crystalline size of the samples. The crystallite size and the strain that were calculated from equation (1) and (2) are tabulated in Table 1. Both the Debye-Scherrer and William-Hall method (UDM) showed that the crystallite size of TiO_2/ZnO bilayer thin film increased after annealing at high temperature indicating it's improvement as a result of the increasing X-RD intensity. Meanwhile, the

reduction of the strain took place after annealing at 600 °C. Similar results have been published by Khan et al.³⁰ and Kheiri et al.²⁷. They showed that the crystallite size and X-RD intensity were directly proportional to the number of ZnO/TiO₂ layer. Therefore, increasing the intensity as a result of the improvement of the crystalline size would reduce stress on the films.

3.2. Surface morphology

The surface morphology of TiO₂/ZnO bilayer thin films was presented by scanning electron microscopy (SEM) image as shown in Fig. 3. The films exhibiting a smooth surface with some small dots which were not well distributed on the surface were observed. Meanwhile, some cracks on the surface of the films were also observed. Furthermore, through SEM, the thickness of TiO₂/ZnO bilayer thin films films that were used to obtain the optical band gap energy was determined. Based on SEM results, the total thickness of the films was 0.98 µm.

3.3 Optical characterization

3.3.1 Transmittance and absorbance

UV-Vis spectroscopy was used to characterize the optical properties of deposited TiO₂, ZnO, and



Fig. 3 — (a) SEM image of TiO₂/ZnO bilayer thin films after annealing at 600 °C, (b) cross section of TiO₂/ZnO bilayer thin films.

	•		•	anatase pł	nase	·	
Annealed samples (TiO ₂ /ZnO)	20	FWHM	(hkl)	Scherer Method ^[31]		William Hall Method ^[32,33] Uniform Deformation Method	
(2)	(deg)	(deg)		Size (nm)	Strain (x10 ⁻³)	Size (nm)	Strain $(x10^{-3})$
	25.41	0.51	(101)*				
	31.86	1.08	(100)				
500 °C	34.49	0.92	(002)	11.63	10.9	6.58	6.4
	36.44	0.68	(101)				
	48.33	0.56	(200)*				
	26.61	0.61	(101)*				
	32.51	0.58	(100)				
600 °C	35.13	0.62	(002)	14.36	7.9	19.30	2.1
	37.09	0.58	(101)				
	48.18	0.68	(200)*				

Table 1 — The structural parameters of bilaver thin films after annealing at various temperatures. Note, Asterisk (*) symbol refers to TiO₂

TiO₂/ZnO bilayer thin films. Fig. 4 depicts the transmittance and absorbance spectra of ZnO and TiO₂ films after annealing at 400 °C and 500 °C, respectively. The transmittance value of TiO₂ film after annealing at 400 °C was around 55 % which was lower compared to the transmittance value after annealing at 500 °C. The transmittance value of TiO₂ after annealing 500 °C was around 85 % in visible regions. Meanwhile, for ZnO film, the transmittance value after annealing 400 °C was around 45 % which increased to 75 % after annealing at 500 °C.

Figure 5 shows the transmittance and absorbance spectra of TiO₂/ZnO bilayer films. The transmittance value of the films was around 75 % - 85 % after annealing at 400 °C. This value decreased to about 60 – 65 % when the film was annealed at 500 °C and increased to 70-75 % after annealing at 600 °C. Furthermore, the absorbance spectra may be divided into three regimes. Firstly, the strong absorbance at $\lambda = 305$ nm, absorbance shoulder region near the band edge for $345 \le \lambda \le 370$ nm and the weak absorbance spectra of TiO₂/ZnO bilayer films also presented some exciton absorption features depending on the



Fig. 4 — The transmittance and absorbance spectra of ZnO and TiO_2 thin films after annealing at 400 °C and 500 °C. The band gap energy of ZnO and TiO_2 thin films are also depicted (inset figure).

annealing temperatures which can be seen in Fig. 5. After annealing at 400, 500 and 600 °C, the exciton absorption showed at 362 nm, 367 nm and 369 nm, respectively, indicating an increase at 600 °C. These values were in close range with the exciton absorption of bulk ZnO which was located at 373 nm³⁴. Ivanova *et al.* have reported the existing of the exciton absorption in the ZnO/TiO₂ deposited film used sol-gel method. The exciton absorption peak was assigned at the wavelength of 362 and 363 nm after annealing at 400 and 500 °C. These numbers were very close to results obtained from this study. The band gap energy of the films was estimated by extrapolation of the equation³⁵.



Fig. 5 — The transmittance and absorbance spectra of TiO_2/ZnO bilayer thin films after annealing at various temperatures from 400 °C to 600 °C for 60 minutes. The inset figure shows the band gap energy of TiO_2/ZnO bilayer thin films before and after annealing with different temperatures from 400 to 600 °C.

$$(\alpha h v)^2 = A (h v - E_g) \qquad \dots (3)$$

$$\alpha = 2.303 x \left(\frac{Abs}{d}\right) \qquad \dots (4)$$

Where α is the absorption coefficient, *Abs* is the optical absorbance, *d* is the thickness of the film, *hv* is the energy and E_g is the band gap energy and *A* is a constant.

Furthermore, the inset Fig. 4, 5 shows the plot of $(\alpha hv)^2$ to photo energy (hv) for TiO₂, ZnO and TiO₂/ZnO bilayer films after annealing at different temperatures from 400 °C to 600 °C for 60 minutes. The band gap energy of the films was obtained by extrapolating the linier part of $(\alpha hv)^2$ against energy (hv). For TiO_2 films, as shown in inset Fig. 4, the band gap energy of TiO₂ films were estimated around 3.89 eV that annealed at 400 °C, which reduced to about 3.77 eV (500 °C annealed). These values were higher than the E_g bulk TiO₂. The large difference between TiO₂ film and TiO₂ bulk was probably due to the grain size and the crystalline structure of thin film and bulk^{36,37}. In addition, The E_g values were slightly larger compared with studies reported by Kheirei et al.27, and Ivanova et al.29. According to Kheiri et al. the estimated value of the optical band gap energy of TiO₂ films were 3.756 eV and 3.677 eV after annealing at 400 °C and 600 °C, respectively. Similar behaviour was reported by Ivanova *et al*, the E_{g} value had 3.74 eV (400 °C) and 3.73 eV (500 °C). However, the results obtained from this study are in agreement with the previous study reported by Quinones et al³⁸. Meanwhile, in the case of ZnO film, the inset Fig. 4 displayed the optical band gap energy of ZnO after annealing at 400 °C and 500 °C. The estimated E_g value for ZnO films were 3.12 eV (400 °C) and 3.25 eV (500 °C). These value were relatively lower than that of previous results^{1,2}.

The band gap energy of TiO₂/ZnO bilayer thin film in inset Fig. 5 has two energy values which are the low and the high energy band gap. These results were quite different when compared with the previous studies that have been carried out in estimating the effective bandgap energy based on derivation from the Touc Plot where only one effective band gap energy was observed in their studies^{27,29}. The first energy (E₁) refers to band to band transition in ZnO films also known as the forbidden gap energy having a value range of 3.29 - 3.32 eV after annealing at 400 °C to 600 °C (Table 2). This energy value referred to the bulk ZnO that have been published in previous studies^{1.2}. In addition, this effective band gap energy was almost similar to the TiO₂ anatase phase of 3.2 eV which was confirmed in the previous studies⁸. The second energy E_2 had a value ranging from 3.56 - 3.64 eV after annealing at 400, 500 and 600 °C, respectively. The E_g value in the combination of TiO₂ and ZnO was lower than the single layer TiO₂ film. The reduction of E_g value in TiO₂/ZnO bilayer thin films was probably due to the penetration of Ti⁺⁴ into ZnO layer that resulted in the reduction in the phase fraction of TiO₂ after it was deposited on top of the ZnO. This created a new binding energy having lower optical band gap energy in comparison to TiO₂ single layer. This explanation was confirmed from the X-RD results where there was a reduction in both X-RD intensity and the plane corresponded to the TiO_2 single layer as a result of the reduction in the phase fraction of TiO₂ layer. Furthermore, similar behaviour was also observed by Kheire et al. where they discovered that reducing the TiO₂ phase fraction increased the density of impurity. In addition, the value of energy E₂ was higher than previous result that has been published by Kheiri et al.27. They reported that the effective band gap energy was in the range of 3.22 – 3.26 eV after annealing at various temperatures from 400 °C to 600 °C which was very close to the pure ZnO. In Kheiri et al, ZnO was the top layer of ZnO/TiO₂/glass. However, the results of effective band gap energy obtained in this study were lower than that reported by Ivanova *et al.*²⁹. They had estimated that the effective band gap energy was in the range of 3.88 - 3.91eV with various annealing temperatures from 400 - 600 °C. Previous study by Hernandes *et al.* reported that the effective band gap energy of Zn₂TiO₂ was 3.7 eV³⁹. The effective band gap energy (E₂) observed in this study was almost similar to that reported by Hernandes.

Table 2 — The I	band gap energy a	nd refractive	index (n) of TiO ₂ ,						
ZnO, TiO ₂ /	ZnO bilayer thin f	ilms derived	from UV-Vis						
spectroscopy.									
Material	Annealing	Eg	Refractive index						
	temperature	(eV)	<i>(n)</i>						
TiO ₂	400 °C	3.89	1.73						
	500 °C	3.77	1.72						
ZnO	400 °C	3.12	1.26						
	500 °C	3.25	2.19						
	400 °C	$E_1 = 3.32$	2.21						
		$E_2 = 3.64$	2.63						
TiO ₂ /ZnO	500 °C	$E_1 = 3.29$	2.36						
		$E_2 = 3.58$	2.94						
	600 °C	$E_1 = 3.30$	2.16						
		$E_2 = 3.56$	2.68						

3.3.2. Refractive index, extinction coefficient and dielectric constant

The other important optical parameter was the refractive index (n) and the extinction coefficient (k) of the film. The refractive index (n) corresponded to the polarizability of ion and local field inside the film. The refractive index (n) and extinction coefficient (k) were estimated using equation⁴⁰.

$$n = \frac{(1+R)+2\sqrt{R}}{(1-R)} \qquad \dots (5)$$

$$R = 1 - (Te^{A})^{2} \qquad \dots (6)$$

$$k = \frac{\alpha \lambda}{4\pi} \qquad \dots (7)$$

Where *R* is the reflectance of the film that is calculated from the transmittance (*T*), *A* is the absorbance, α is the absorption coefficient and λ is the wavelength. The refractive index (*n*) and the extinction coefficient (*k*) as a function of wavelength for single layer TiO₂, ZnO and TiO₂/ZnO bilayer thin films were displayed in Fig. 6, 7. In Fig. 6, the refractive index (*n*) of TiO₂ annealed at 400 °C was higher than that annealed at 500 °C. In addition, the refractive index (*n*) reduced significantly from 2.6 (λ =300 nm) to 1.1 (λ =412 nm) after annealed at 500 °C and became constant at λ >425 nm. For ZnO



Fig. 6 — The refractive index (*n*) and the extinction coefficient (*k*) of TiO_2 and ZnO after annealing at 500 °C and 600 °C for 60 minutes.

films, similar behaviour in TiO₂ was observed in ZnO film where the refractive index (n) tended to be constant at λ >450 nm. Meanwhile, for the extinction coefficient (k), both in TiO₂ and ZnO were lower than that of the refractive index (n). For TiO_2/ZnO bilayer thin films (see Fig. 7), the refractive index (n) was in the range of 2.6 - 3.0 at a wavelength of less than 360 nm. Subsequently, it reduced quite significantly to 1.36, 1.56 and 1.27 at wavelength of 394 nm (3.15 eV) after annealing at 400 °C, 500 °C and 600 °C, respectively. The effective band gap energy at 3.30 eV and 3.56 eV were assigned to the refractive index (n) of 2.16 and 2.68 (See Table 2). In comparison with the previous studies⁴¹, the refractive index (n) of polycrstalline ZnO film and TiO₂ were assigned 2.22 and 2.32, respectively. Ivanova et al. showed that the refractive index (n) of ZnO/TiO_2 was 2.19 which was lower than results obtained from this study. Meanwhile, the extinction coefficient (k) was lower than that of the refractive indices for all films after annealing at 400 °C, 500 °C and 600 °C, respectively.

Furthermore, the fundamental excitation spectrum of the films was described by means of a frequency dependent on the complex electronic dielectric constant which was defined as $\varepsilon = \varepsilon_r + i\varepsilon_i$, where ε_r and ε_i are the real and imaginary part of dielectric constant, respectively that corresponded to the refractive index *n* and the extinction coefficient *k* according to equation⁴².

$$\varepsilon_r = n^2 - k^2 \qquad \dots (8)$$

$$\varepsilon_i = 2nk$$
 ... (9)

Figure 8 shows the real ε_r and imaginary part ε_i of dielectric constant of bilayer TiO₂/ZnO bilayer thin films after annealing at 400 °C to 600 °C for 60



Fig. 7 — The refractive index (*n*) and the extinction coefficient (*k*) of TiO_2/ZnO after annealing at various temperatures from 400 °C to 600 °C for 60 minutes.



Fig. 8 — The real (ϵ_r) and imaginary (ϵ_i) part of dielectric constant of TiO₂/ZnO bilayer thin films after annealing at various temperatures from 400 °C to 600 °C for 60 minutes.

minutes. The variation of real and imaginary part of dielectric constant was inversely proportional to the wavelength, reaching a maximum at the wavelength of around 362 nm and vice versa. In addition, the imaginary part of dielectric constant was similar in behaviour to the real dielectric constant which shows that there was a maximum dielectric constant with a certain value of wavelength.

4 Conclusion

Thin films of metal oxide TiO₂, ZnO and bilayer TiO₂/ZnO bilayer thin films have been successfully synthesized and characterized using X-ray diffraction, SEM and UV-Vis spectroscopy measurements after annealing at different temperature in the range 400 °C - 600 °C. From X-RD results obtained, it was observed that the X-RD peak intensity in TiO₂/ZnO was more pronounced in the films after annealing at 600 °C compared to the films annealed at 400 and 500 °C, respectively. Furthermore, the optical properties of TiO₂/ZnO bilayer thin films showed a high value transmittance which was around 80% in the visible regions after annealing at 600 °C. The effective optical band gap energy of mixed TiO₂ and ZnO which was derived from Touch plot gave two regions at low and high energy. In addition, the optical band gap energy of the TiO₂ layer was higher than that of the mixed TiO₂ and ZnO layers. The refractive index (n), the extinction coefficient (k) and the dielectric constant showed similar behaviour in the single layer and mixed layer. Based on the optical characterization of TiO₂, ZnO and TiO₂/ZnO bilayer, these materials were suitable for application in optoelectronic devices.

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