

First Principle Study of Electronic, Optical and Thermoelectric Properties of CuInS_2 and CuInSe_2

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We report the bandgap, thermoelectric and optical properties of CuInS_2 and CuInSe_2 ternary chalcopyrite compounds based on DFT calculations. Our calculations shows that both CuInS_2 and CuInSe_2 have a direct bandgap which were at the Γ -points. The computed bandgap were 1.35 and 0.85 eV for CuInS_2 and CuInSe_2 . The optical properties analysis shows that the fundamental edge of absorption arise at 0.82 eV and 0.35 eV along the perpendicular and parallel polarization for CuInS_2 , while it arise at 0.13 eV and 0.16 eV along the perpendicular and parallel polarization for CuInSe_2 . The static dielectric constant, static refractive index and birefringence were then calculated. The calculated birefringence was negative, which meets the non-critical phase matching (NCPM) requirement, which is beneficial for high-performing laser systems. The optical absorption threshold lies at 1.4 and 0.83 eV for CuInS_2 and CuInSe_2 . These compounds show low reflectivity and high absorption in the visible region. Both compounds have high electrical conductivity and Seebeck coefficient, making them promising candidates for thermoelectric devices.

Keywords: Seebeck coefficient; DFT; Absorption coefficient

1 Introduction

Ternary I-III-VI₂ compounds are semiconducting compounds that crystallize into chalcopyrite structures. They have attracted considerable attention because of their excellent absorption coefficients and desirable direct bandgaps for applications in photovoltaics. I-III-VI₂ compounds are promising candidates for applications in solar cells¹⁻³, light-emitting diodes^{4,5} and nonlinear optics^{6,7}. Cu-based I-III-VI₂ compounds are especially attractive because of their nontoxicity and high conversion efficiency in solar cells in contrast to lead-based solar cells, which are toxic and not environment friendly. For the purpose of this study, we consider CuInS_2 and CuInSe_2 because of their technological applications. The direct bandgap and non-cubic structure of CuInS_2 lead to interesting electrical and optical properties, including third-order nonlinear optical properties⁸⁻¹⁰. Experimental synthesis of CuInS_2 was performed using different methods, and its properties were studied^{11,12}. Theoretical studies on the band structure and lattice dynamics of CuInS_2 were performed using pseudo potential DFT¹³. The influence of P and In doping on the band structure and stability of CuInS_2 has been studied¹⁴. The structural,

dynamical, and dielectric properties of CuInS_2 were studied using local density approximation¹⁵. Theoretical studies of the thermodynamic and elastic properties of CuInS_2 have been performed using DFT¹⁶. CuInSe_2 is extensively used in photovoltaic materials because of its large absorption coefficient¹⁷, high conversion efficiency¹⁸, and low production cost¹⁹. Various experimental studies have been performed to synthesize CuInSe_2 and better understand its electrical and optical properties at ambient pressure²⁰⁻²⁷. The structure of CuInSe_2 changes upon application of pressure, which increases its absorption coefficient²⁸. The transport properties of CuInSe_2 and CuInS_2 are also important parameters that significantly influence its practical applications²⁹.

2 Computational Details

Calculations were performed within the DFT framework using FP-LAPW as implemented in WIEN 2k³⁰. The MBJ³¹ potential was used as the exchange-correlation potential to compute the band structure and optical properties. The structures of CuInS_2 and CuInSe_2 were optimized to obtain stable structures using GGA³². For good energy convergence, the cut-off energy was set at -6.0 Ry, 3000 k-points were used, and the cut off plane wave vector in the interstitial region was $R_{\text{MT}} \times K_{\text{max}} = 8$. A maximum angular momentum quantum value of 10, and Fourier

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charge density of 12Ry were used. The energy and charge convergences were set as 0.0001Ry and 0.001e, respectively. The thermoelectric properties were calculated using semi-classical Boltzmann transport theory with a constant τ approximation as implemented in Boltz TraP2³³.

3 Results and Discussion

3.1 Electronic properties

The computed band structures of CuInS_2 and CuInSe_2 are shown in Fig. 1 (a) and (b). They were computed across the sharp symmetry points of the BZ from Γ to H, N, Γ , and P. The top VB and CB minima

were located at the Γ -points of the BZ for CuInS_2 and CuInSe_2 , indicating that both have a direct bandgaps. Fig. 2 shows the computed TDOS and PDOS for better insight into the energy bands. Three sub-bands emerged from the valence bands for both compounds. The upper sub-bands for CuInS_2 lies between the fermi energy and -1.7 eV and the lower two sub-bands lies between -3.4 to -5.1 eV and -5.7 to -6.1 eV respectively. The upper sub-band of CuInSe_2 lies between the fermi energy and -1.4 eV and the lower two sub-bands lies between -3 to -4.8 eV and -5.4 to -6.3 eV respectively. The highest orbital occupied in the VB was characterized by Cu 3d, S 3p, and Se 4p

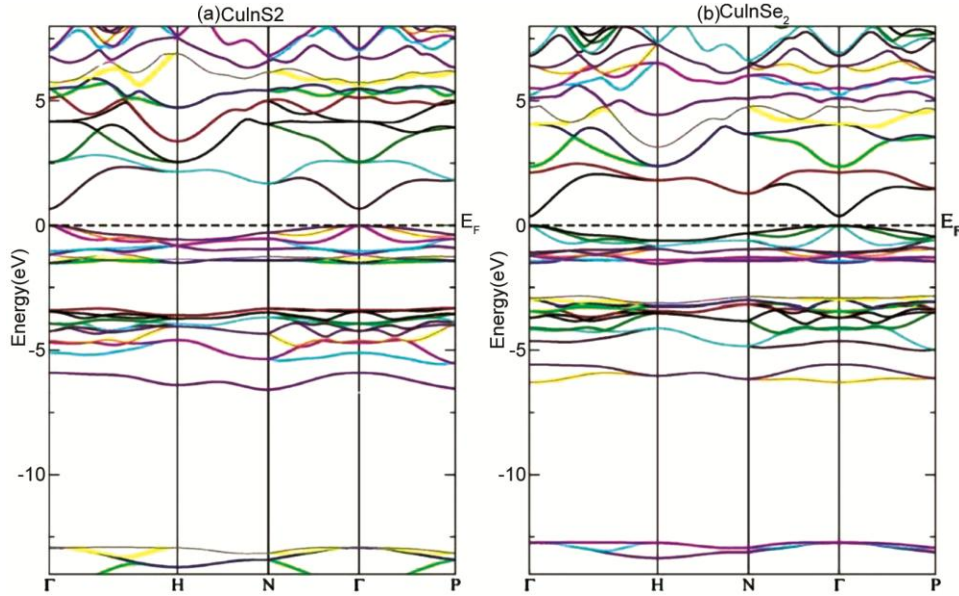


Fig. 1 — Calculated Electronic Band Structure of (a) CuInS_2 and (b) CuInSe_2 .

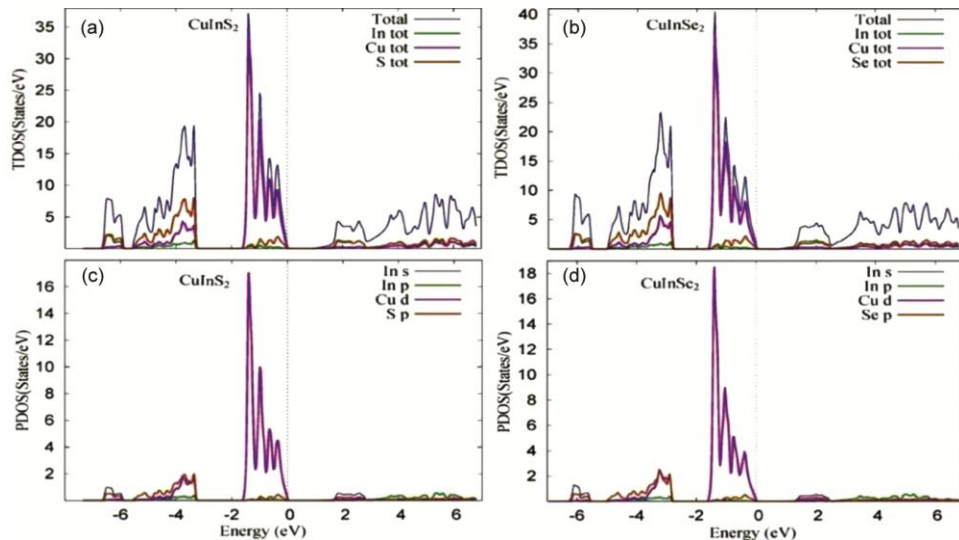


Fig. 2 — Calculated Total and Partial Density of States for CuInS_2 and CuInSe_2 .

states whereas the lower CB was mostly attributed to the In 5s states for CuInS₂ and CuInSe₂. The upper VB signifies an anti bonding of Se 4p and S 3p states with Cu 3d states, while the CB minima is mostly attributed to the anti bonding of In 5s, Se 4p with S 3p states. The calculated bandgaps is shown in Table 1. The computed bandgap is less than the experimental result indicating the inadequacy of DFT approach for calculating the band structure.

3.2. Optical properties

The linear response of a macroscopic optical properties of a solids are usually described by means of complex dielectric function^{34,35} given by

$$\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega) \quad \dots (1)$$

$\varepsilon_1(\omega)$ and $\varepsilon_2(\omega)$ are dispersive and absorptive parts of the complex dielectric function. The imaginary part $\varepsilon_2(\omega)$ of dielectric function with appropriate momentum matrix is given by³⁶

$$\varepsilon_2(\omega) = \frac{e\hbar}{\pi m^2 \omega^2} \sum_{v,c} \int |M_{cv}(k)|^2 \delta[\omega_{cv}(k) - \omega] d^3k \quad \dots (2)$$

$\varepsilon_1(\omega)$ can be obtained from $\varepsilon_2(\omega)$ using the Kramer-Kronig transformation³⁷ as,

$$\varepsilon_1(\omega) = 1 + \frac{2}{\pi} P \int_0^{\infty} \frac{\omega' \varepsilon_2(\omega')}{\omega'^2 - \omega^2} d\omega' \quad \dots (3)$$

The other optical parameters such as, reflectivity³⁸, absorption coefficient (α)³⁸, refractive index³⁹ and extinction coefficient³⁹ can be calculated as follows,

$$R(\omega) = \left| \frac{\varepsilon_2^2(\omega) - 1}{\varepsilon_2^2(\omega) + 1} \right| \quad \dots (4)$$

$$\alpha(\omega) = \sqrt{2\omega} \left[\left(\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)} - \varepsilon_1(\omega) \right)^{\frac{1}{2}} \right] \quad \dots (5)$$

$$n(\omega) = \left[\frac{\varepsilon_1}{2} + \frac{\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)}}{2} \right]^{\frac{1}{2}} \quad \dots (6)$$

$$k(\omega) = \left[\frac{\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)}}{2} - \frac{\varepsilon_1}{2} \right]^{\frac{1}{2}} \quad \dots (7)$$

Figure 3 (a) shows the calculated real part of the dielectric function for CuInS₂ and CuInSe₂. The overall structure of $\varepsilon_1(\omega)$ is similar for both CuInS₂ and CuInSe₂. There are four main peak located at 0.8, 2, 4.1, and 5.2 eV for CuInS₂ and four main peaks located at 0.5 eV, 1.6 eV, 4.23 eV and 6.3 eV for

Table 1 — Calculated Bandgap

Compounds	E _g (eV)		
	Our work	Other Theoretical work	Experimental work
CuInS ₂	1.35	0.26 ^a , 0.873 ^b , 1.37 ^b	1.55 ^c , 1.53 ^d
CuInSe ₂	0.85	0.26 ^a , 0.748 ^b , 1.23 ^b	1.04 ^d

^aRef. [44], ^bRef. [45], ^cRef [46], ^dRef [47]

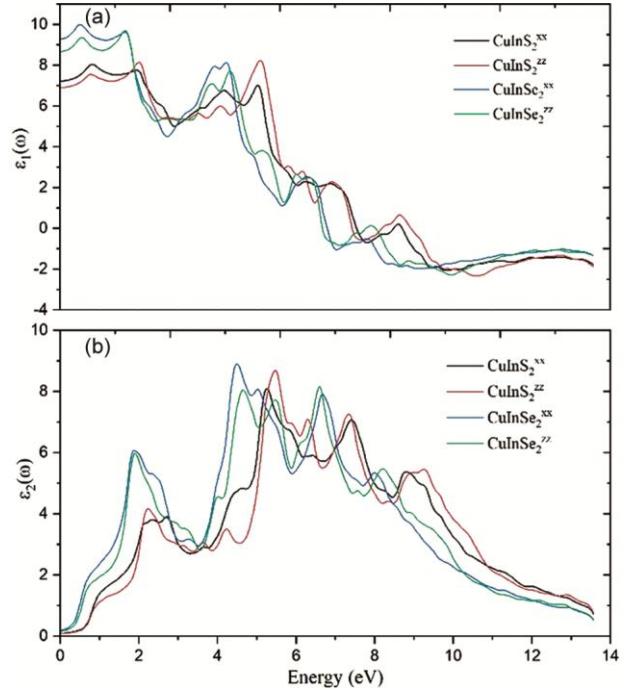


Fig. 3 — Calculated (a) Real and (b) Imaginary parts of Dielectric Function for CuInS₂ and CuInSe₂.

CuInSe₂. These peaks originate from the electronic transitions between the upper VB and the lower CB, and the peak intensity decreases with increasing energy. The energy at which $\varepsilon_1(\omega)$ is zero in the ultraviolet region indicates the disappearance of the dispersion that causes absorption. The negative value of $\varepsilon_1(\omega)$ indicates the non-propagation of electromagnetic waves in this region and can be used as a shield against electromagnetic waves in this region. The static dielectric constant has an inverse relationship with the bandgap, which can be explained using the Penn model⁴⁰. The calculated static dielectric constants are listed in Table 2. The calculated spectra of $\varepsilon_2(\omega)$ for CuInS₂ and CuInSe₂ are presented in Fig. 3 (b). The fundamental edge of absorption arise at 0.82 eV and 0.35 eV along the perpendicular and parallel polarization for CuInS₂, while it arise at 0.13 eV and 0.16 eV along the perpendicular and parallel polarization for CuInSe₂. The fundamental edge corresponds to a transition

Table 2 — Calculated static dielectric constant

Compounds	Our Work		Other Work	
	$\epsilon^{xx}(0)$	$\epsilon^{zz}(0)$	$\epsilon^{xx}(0)$	$\epsilon^{zz}(0)$
CuInS ₂	7.23	6.91	7.51 ^a , 8.6 ^b , 9.71 ^c	6.97a, 8.4 ^b , 9.68 ^c
CuInSe ₂	9.29	8.67	9.86 ^a , 11 ^b	8.95 ^a , 10.8 ^b

^aRef^[48], ^bRef^[49], ^cRef^[15]

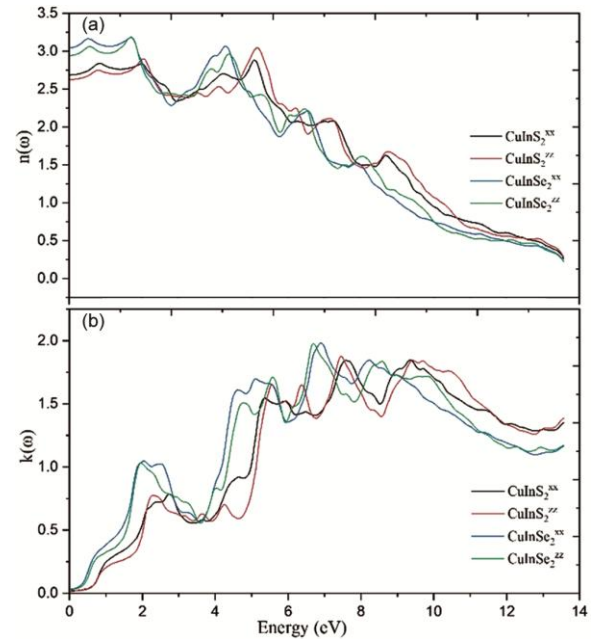
Table 3 — Calculated static refractive index with birefringence

Compounds	Our Work			Other Work		
	$n^{xx}(0)$	$n^{zz}(0)$	$\Delta n(0)$	$n^{xx}(0)$	$n^{zz}(0)$	$\Delta n(0)$
CuInS ₂	2.69	2.62	-0.07	2.76 ^a ,	2.6 ^a ,	-0.16 ^a ,
CuInSe ₂	3.04	2.94	-0.1	3.17 ^a ,	3.01 ^a ,	-0.16 ^a ,

^aRef^[48]

between the lowest CB and highest VB. Beyond the threshold energy, three main peaks are located at 2.74, 5.26, and 7.41 eV for CuInS₂. These peaks originate from the transitions of the hybridized bonding of S 3p and Cu 3d states. Beyond the threshold energy three main peaks are located at 1.86, 4.5 and 6.6 eV for CuInSe₂. These peaks are mainly attributed to the transitions of hybridized bonding of Se 4p and Cu 3d states. These peaks correspond to region of strongest absorption of, and the first peak lies in the visible region and extends all the way to the ultraviolet region. Peaks with lower amplitude arise owing to the nature of hybridization of π and σ bonding and the anti-bonding of Cu 3d states and the S 3p and Se 4p states.

Figure 4 (a) shows the calculated refractive index for CuInS₂ and CuInSe₂. The refractive index describe the macroscopic polarization of a material due to an incident electromagnetic waves. Two main peaks were observed in the refractive index spectra. The first peak lies in the infrared region, whereas the second peak lies in the ultraviolet region. $n(\omega)$ exhibits anisotropy that results in birefringence, which accounts for the prominent nonlinear optical properties of these compounds. The extent of birefringence, which is given by $\Delta n = n^{\parallel} - n^{\perp}$, along with the static refractive index, is listed in Table 3. $\Delta n(0)$ is negative for both compounds, which meets the non-critical phase matching (NCPM) requirement, which is beneficial for high-performing laser systems. Figure 4 (b) shows the calculated extinction coefficient for CuInS₂ and CuInSe₂. The extinction coefficient spectra of CuInS₂ and CuInSe₂ are almost identical because of the similarity in their band structures. Both compounds has intense peak between 4.8 eV to 8.5 eV. The intensity of the peak exhibits a red shift and it decrease with increase in energy. This indicates that the electric field vanished

Fig. 4 — Calculated (a) Refractive index and (b) Extinction coefficient for CuInS₂ and CuInSe₂.

with respect to the incident energy. $k(\omega)$ represents damping of the oscillating electric field amplitude. Three sharp peaks appears around 5.34 eV, 7.5 eV and 9.34 eV for CuInS₂ and at 5.1 eV, 6.8 eV and 8.2 eV for CuInSe₂, indicating region of strongest absorption. The extinction coefficient decreased after reaching a peak owing to low absorption. The correlation between the absorption and extinction coefficients can be explained using Beer-Lambert law⁴¹.

Figure 5 (a) shows the calculated absorption coefficients of CuInS₂ and CuInSe₂. Optical absorption is an important optical parameter that displays the luminosity arising from electronic transitions. The absorption coefficient provides information on the number of electromagnetic waves with distinct

energies that enter a material prior to absorption. The incident photons may be absorbed through direct or indirect transitions, depending on whether the energy gap is lower than the incident radiation energy. The absorption threshold lies at 1.4 eV and 0.83 eV for CuInS₂ and CuInSe₂ which are in good agreement with other calculation at 1.5eV⁴² for CuInS₂ and 0.83 eV⁴³ for CuInSe₂. The absorption has maxima of 175.7 x 10⁴ and 186.6 x 10⁴ cm⁻¹ located at 9.4 eV and 10.4 eV along the perpendicular and parallel directions for CuInS₂ while the absorption has

maxima of 157.8 x 10⁴ and 171.8 x 10⁴ cm⁻¹ located at 9.1 and 9.8 eV along the perpendicular and parallel directions for CuInSe₂. The absorption coefficient strongly increases with increasing energy, indicating the beginning of a direct transition. The reflectivity spectra of CuInS₂ and CuInSe₂ are shown in Fig. 5 (b). The reflectivity intensify along the infrared region and then decrease as it approach the visible region. The static reflectivity was 20% and 21 % along the perpendicular and parallel polarizations for CuInS₂ while the static reflectivity for CuInSe₂ was 26 % and 24% along the perpendicular and parallel polarizations. The reflectivity exhibited the highest intensity in the ultraviolet region for both compounds, which can be utilized to create Bragg's reflectors.

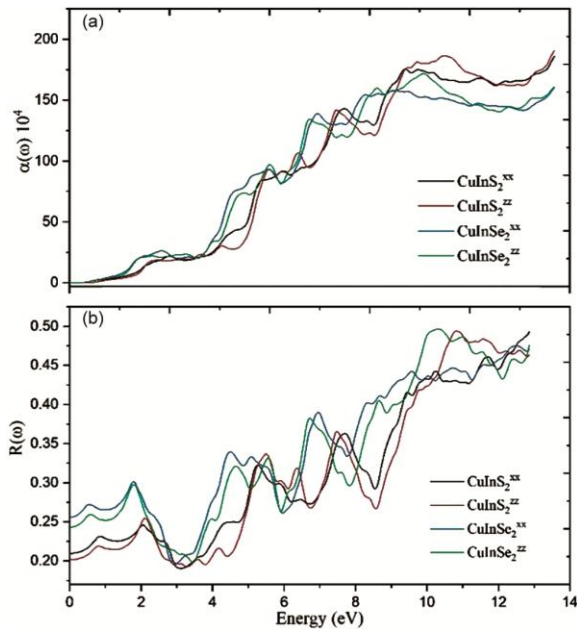


Fig. 5 — Calculated (a) Adsorption coefficient and (b) Reflectivity for CuInS₂ and CuInSe₂.

3.3. Thermoelectric properties

The thermoelectric properties of CuInS₂ and CuInSe₂ were calculated using semi-classical Boltzmann transport theory, as implemented in BoltzTraP2³³. τ (relaxation time) was assumed to be constant. The thermal conductivity due to lattice vibrations was not considered when calculating ZT. The calculated electrical conductivity is shown in Fig. 6 (b). The increase in temperature causes easier movement of the carriers from the VB to the CB, resulting in an increase in σ/τ with temperature. The calculated $(\sigma/\tau)^{xx}$ and $(\sigma/\tau)^{zz}$ for CuInS₂ at room temperature were 8.55 x 10¹⁹ (Ωms)⁻¹ and 6.84 x 10¹⁹ (Ωms)⁻¹. For CuInSe₂, the calculated $(\sigma/\tau)^{xx}$ and $(\sigma/\tau)^{zz}$ values at room temperature were 7.76 x 10¹⁹ (Ωms)⁻¹ and 5.74 x 10¹⁹ (Ωms)⁻¹. The Seebeck coefficient is a measure of the magnitude of change in induced

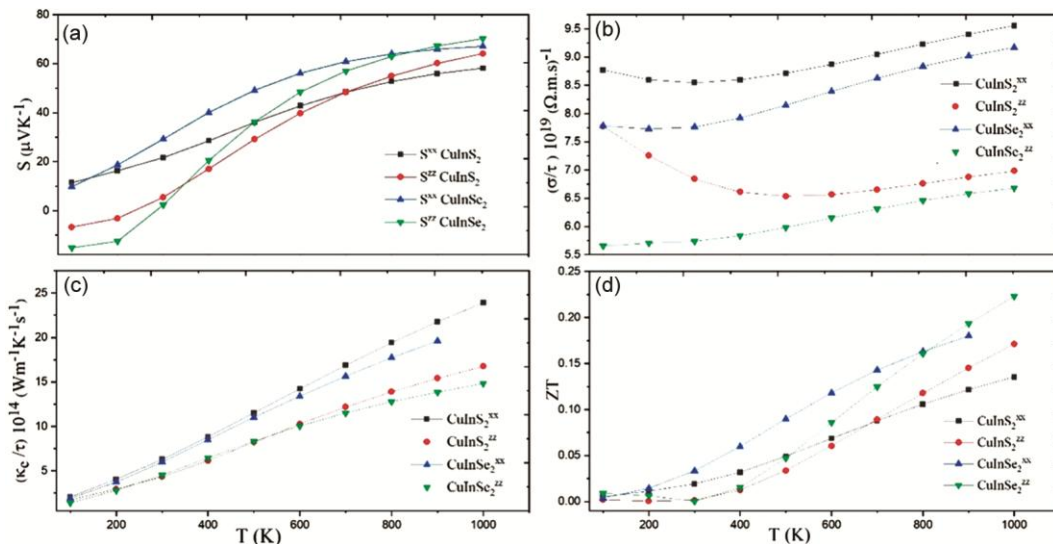


Fig. 6 — Calculated (a) Seebeck coefficient (b) Electrical conductivity (c) Thermal conductivity and (d) Figure of merit for CuInS₂ and CuInSe₂.

voltage with respect to the change in temperature, given by $S = \Delta V / \Delta T$ ⁵⁰. Fig. 6 (a) shows the variations of S (Seebeck coefficient) with temperature. The Seebeck coefficient increased linearly as the temperature increase 100 K to 1000 K for both compounds. The calculated S^{xx} and S^{zz} for CuInS₂ at room temperature were 21.7 μVK^{-1} and 5.43 μVK^{-1} respectively. For CuInSe₂ the calculated S^{xx} and S^{zz} at room temperature were 29.32 μVK^{-1} and 2.32 μVK^{-1} . A positive Seebeck coefficient indicated the presence of p-type carriers. Both compounds have high electrical conductivity with a high Seebeck coefficient at room temperature, which is characteristic of good thermoelectric materials.

Thermal conductivity is attributed to electronic and lattice vibrations, which constitute the total thermal current. The contribution of the lattice vibrations was ignored in the calculations. The calculated thermal conductivities are shown in Fig. 6 (c). The calculated $(\kappa_e/\tau)^{xx}$ and $(\kappa_e/\tau)^{zz}$ values for CuInS₂ at room temperature were 6.29×10^{14} and $4.32 \times 10^{14} \text{Wm}^{-1}\text{K}^{-1}\text{s}^{-1}$, respectively. For CuInSe₂ the calculated $(\kappa_e/\tau)^{xx}$ and $(\kappa_e/\tau)^{zz}$ at room temperature were 5.9×10^{14} and $4.5 \times 10^{14} \text{Wm}^{-1}\text{K}^{-1}\text{s}^{-1}$. The relationship between electrical conductivity and thermal conductivity is given by the Wiedemann-Franz law $LT = \kappa/\sigma$ ⁵¹. To obtain good thermoelectric materials, this ratio must be small. The κ/σ ratios for CuInS₂ and CuInSe₂ were quite small, of the order 10^{-5} , making them good candidates for thermoelectric devices. The efficiency of the thermoelectric material was assessed using the dimensionless figure of merit $ZT = S^2\sigma T/\kappa$ ⁵². Ideally thermoelectric materials should exhibit ZT values near or greater than unity. The variation of ZT with temperature is shown in Fig. 6 (d). It increased linearly with temperature. The value of ZT at room temperature were 0.019 and 0.034 for CuInS₂ and CuInSe₂ respectively. The calculated ZT at room temperature is quite low, although the Seebeck coefficient and σ/τ are high. This was owing to the exclusion of lattice thermal conductivity for calculating the figure of merit.

4 Conclusions

We investigate bandgap, optical parameters, and thermoelectric properties of CuInS₂ and CuInSe₂ ternary chalcopyrite compounds based on DFT calculations. Three sub-bands emerged from the valence bands for both compounds. The upper sub-bands for CuInS₂ and CuInSe₂ lie between the fermi energy and -1.7 eV and -1.4 eV respectively. Our

calculations shows that both CuInS₂ and CuInSe₂ have a direct bandgap which were at the Γ -points. The computed bandgaps were of 1.35 eV and 0.85 eV for CuInS₂ and CuInSe₂. The calculated bandgap is less than the experimental result indicating the inadequacy of DFT approach for calculating the energy bands. The real part of the dielectric function of CuInS₂ and CuInSe₂ has four main peak located at 0.8, 2, 4.1, and 5.2 eV for CuInS₂ and the main peak for CuInSe₂ are located at 0.5, 1.6, 4.23, and 6.3 eV. The calculated $\epsilon(0)$ and $n(0)$ were compared with available experimental and theoretical data and are in reasonable agreement. The fundamental edge of absorption arise at 0.82 eV and 0.35 eV along the perpendicular and parallel polarization for CuInS₂, while it arise at 0.13 eV and 0.16 eV along the perpendicular and parallel polarization for CuInSe₂. $\Delta n(0)$ was negative for both compounds, which meets the non-critical phase matching (NCPM) requirement, which is beneficial for high-performing laser systems. The extinction coefficient exhibits three sharp peaks appears around 5.34 eV, 7.5 eV and 9.34 eV for CuInS₂ and at 5.1 eV, 6.8 eV and 8.2 eV for CuInSe₂, indicating region of strongest absorption. The absorption threshold lies at 1.4 and 0.83 eV for CuInS₂ and CuInSe₂. Both compounds exhibit high Seebeck coefficient and electrical conductivity, making them promising candidates for thermoelectric devices.

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